Structural Study on Iron Oxide Nanoparticles Prepared by Sol-Gel Method

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Abstract—Fine nanosized metal oxide namely, iron oxide (α -Fe₂O₃) has been synthesized by sol-gel method using Ferric Chloride as the starting precursor. X-ray diffraction (XRD) study has been conducted to identify the polymorph of the prepared iron oxide nanoparticles. From the study it was observed that the prepared nanoparticles exhibited rhombohedral α -Fe₂O₃ phase without any other phases magnetite (Fe₃O₄) or a mixture of magnetite (Fe₃O₄) and maghemite (γ -Fe₂O₃). All the structural parameters such as lattice constants, unit cell volume, density, crystalline size, micro strain, dislocation density, texture co-efficient were calculated from the XRD results.

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Index Term — nanoparticle; Fe₂O₃; sol-gel; XRD; hematite; rhombohedral;maghemite

1 INTRODUCTION

ransition metal oxides have many applications as catalysts, sensors, superconductors, and many bio-medical applications. Metal oxides constitute an important class of materials that are involved in environmental science, electrochemistry, biology, chemical sensors, magnetism, and other fields. One of the most important applications is biomedical applications. Iron oxide belongs to the most abundant minerals and occurs with a large variety of stoichiometries, structures, and properties. Iron oxide exists in three forms in nature: magnetite (Fe3O4), maghemite (y-Fe2O3), and hematite (α -Fe2O3). Hematite is the oldest known of the iron oxides and is widespread in rocks and soils. It is also known as ferric oxide, iron sesquioxide, red ochre, specularite, specular iron ore, kidney ore, or martite. Hematite is blood-red in color if finely divided, and black or grey if coarsely crystalline. It is extremely stable at ambient conditions, and often is the end product of the transformation of other iron oxides. Magnetite is also known as black iron oxide, magnetic iron ore, loadstone, ferrous ferrite, or Hercules stone. It exhibits the strongest magnetism of any transition metal oxide. Maghemite occurs in soils as a weathering product of magnetite, or as a product of heating of other iron oxides. It is metastable with respect to hematite, and forms continuous solid solutions with magnetite. The important features of Hematite (a-Fe2O3) is its density

(5.312 gm/cc), and it is weakly ferromagnetic or antiferromagnetic, melting point is about 1350 °C, and the crystallographic system is rhombohedral or hexagonal. By modifying the growth conditions, the size of the iron oxide particles can be reduced to nanosize. The crystal structure of the technologically important Hematite is shown in Fig. 1.

The iron atom has a strong magnetic moment due to four unpaired electrons in its 3d orbitals. When crystals are formed from iron atoms, different magnetic states can arise. In the paramagnetic state, the individual atomic magnetic moments are randomly aligned with respect to each other, and the crystal has a zero net magnetic moment. If this crystal is subjected to an external magnetic field, some of these moments will align, and the crystal will attain a small net magnetic moment. In a ferromagnetic crystal, all the individual moments are aligned even without an external field.

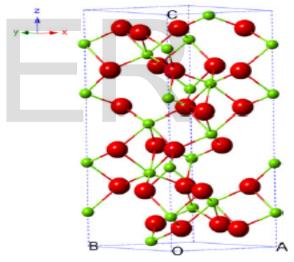


Fig. 1 Crystal structure and crystallographic data of hematite (green ball is Fe3+ and the red ball is O2-)

ferromagnetic crystal, on the other hand, has a net magnetic moment from two types of atoms with moments of different strengths that are arranged in an antiparallel fashion. If the antiparallel magnetic moments are of the same magnitude, then the crystal is antiferromagnetic and possesses no net magnetic moment. Hematite is paramagnetic at temperatures above its Curie temperature of 956 K. At room temperature, it is weakly ferromagnetic and undergoes a phase transition at 260 K (the Morin temperature, TM) to an antiferromagnetic. The magnetic behavior of hematite depends on crystallinity, particle size and on the extent of cation substitution. The Morin temperature of hematite decreases as the particle size decreases and tends to vanish for particles smaller than 8 - 20

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nm. Poor crystallinity and substitution of cations tend to lower TC and TM (and may even completely suppress the Morin transition) at all temperatures.

Since iron oxide is a technology important material, a systematic study has been initiated to prepare it through solgel method using ferric chloride as the precursor and ammonium hydroxide as the reducing agent. Prepared nanoparticles have been analyzed to evaluate its structural properties from X-ray diffraction studies. Structural parameters such as lattice parameter, unit cell volume, density, crystalline size, micro strain, dislocation density, texture co-efficient were all extracted from the XRD data.

2 EXPERIMENTAL

Iron oxide nanoparticles were synthesized using sol-gel method. In sol-gel method, there are two types of materials or components, "sol" and "gel". Sols are solid particles in a liquid subclass of colloids and gels are ligands contained in liquid. This method can produce highly pure and well controlled nanoparticle. It is a low temperature, less energy consumption and less pollution process. This process involves formation of sols in a liquid and it is reduced to the desired product using a reducing agent. In the present study Ferric Chloride is used as the precursor, Ethanol as the solvent and Ammonium Hydroxide as the reducing agent. Initially, Ferric Chloride is dissolved in the solvent and the reducing agent is added drop by drop to produce the iron oxide nanoparticles. The Ethanol and Ammonium Hydroxide is added in 4:1 radio for a makeup of 0.5M (200ml) solution. Prepared sample is then washed repeatedly using water and ethanol in-order to remove the impurities. Then they were dried to get the Nano powder. The yield was about 19.520gms. Then half of the sample was calcinated at 300°C for 3hrs. and the sample is hence forth mentioned as FeO-1 and the remaining half was calcinated at 800°C for 3hrs. and is named as FeO-2.

3 RESULTS AND DISCUSSION

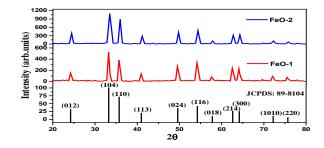


Fig. 2 X-Ray spectra of the synthesized iron oxide nanoparticles.

Fig. 2 shows the X-ray diffraction spectra of the synthesized iron oxide nanoparticles (FeO-1 & FeO-2).

The synthesized iron oxide nanopowders were characterized using X-ray diffraction. XRD pattern indicates that the prepared iron oxide was in α -Fe2O3 phase exhibiting

rhombohedral structure. Observed peaks are in defined positions that shows the formation of α -Fe2O3 without any impurity peaks of any other phase of iron oxide, which indicates a high degree of purity of the prepared samples. The broadening of the X-ray diffraction lines reflects the nanoparticle nature of the sample. In XRD, all the peaks are indexed and the d-values are compared with the JCPDS standards [JCPDS file no. 89-8104]. Using the XRD data all other related structural parameters have been calculated.

The unit cell edges was calculated using the equation:

$$\frac{1}{d^2} = \frac{4}{3} \left(\frac{h^2 + hk + k^2}{a^2} \right) + \frac{l^2}{c^2}$$

where, 'd' is the inter planner distance, 'a', 'c' are the lattice parameters and 'h, k, l' are the Miller indices of lattice planes. The unit cell volume was calculated using the equation;

unit cell volume,
$$V = \frac{\sqrt{3} a^2 c}{2} (Å^3)$$

where 'a','c' are the lattice parameters. The density was calculated using;

$$Density, \rho = \frac{FW * Z * 1.66}{Unit \ cell \ volume} \ (gm/cc^3)$$

where, 'FW' is the formula weight which is 159.69 for Fe2O3 Z = 6 (No. of units of Fe2O3 in a unit cell) The crystalline size was calculated using:

$$D = \frac{0.97 * \lambda}{\beta_{2\theta} * \cos\theta} (nm)$$

where; $\lambda = 1.5462 \text{ Å}$, $\beta 2\theta = FWHM$ in radian

The micro strain was calculated using:

$$\varepsilon = \frac{\beta_{2\theta} * \cos\theta}{4}$$

where; $\beta 2\theta$ = FWHM in radian

The dislocation density was calculated using:

$$\delta = \frac{1}{D^2} \ (lines/m^2)$$

where; D is the crystalline size

The texture co-efficient was calculated using:

International Journal of Scientific & Engineering Research Volume 9, Issue 7, July-2018 ISSN 2229-5518

$$TC = \frac{I_{hkl}/I_{O_{hkl}}}{\frac{1}{N * \sum_{N} I_{hkl}}/I_{O_{hkl}}}$$

where; 'I' is the intensity and 'N' is the total number of peaks.

 TABLE I. LATTICE PARAMETER OF THE PREPARED A-FE2O3

 NANOPARTICLE

SAMPLE	LATTICE PARAMETER (Å)		UNIT CELL VOLUME (Å3)		
DETAILS	EXPERIMENT AL	Standa RD	EXPERIMENTA L	Standar D	
FEO-1	A = 4.99500 C=13.62666	A= 5.023 C=13.70	294.436086	299.52	
FEO-2	A = 5.03170 C=13.73443	8	301.14159		

In FeO-1 the lattice parameter value is less compared to the standard lattice parameter, this is because of the preparative parameters. As the sample FeO-1 was prepared at low temperature, many of the lattice sites are vacant. Hence the lattice length both in 'a' and 'c'-axis is less compared to the standard bulk sample. However, in FeO-2, the sample calcined at 800°C, all the lattice sites are completely filled and hence the lattice parameter value is nearly equal to the standard lattice parameter value. Since the lattice parameter is less in FeO-1, the unit cell volume is less compared to the standard unit cell volume. Since every lattice locations are filled with Fe and O ions in FeO-2, the unit cell volume is increased and it is nearly equal to the standard unit cell volume.

TABLE II. RELATED STRUCTURAL PARAMETER OF A-FE2O3 NANOPARTICLE

SAMP	DENSITY (GM/CC3)				DISLOCATI
LE		Standa	CRYSTALLI NE SIZE	MICRO STRAIN	ON DENSITY
DETAI LS	Exp.	RD	(D) NM		(*10^15)
20					LINES/M2
FEO- 1	5.4018 9	5.312	22.71590	0.001 62	1.93793
FEO- 2	5.2816 09		24.17427	0.001 52	1.7117

As the volume decreases, the density increases and vice versa. Hence as the unit cell volume of FeO-1 is less the density value is increased and it is high compared to its standard density value. Since the unit cell volume of FeO-2 is high or nearly equal to the standard value the density value is reduced and it is less than the standard density value. As the temperature increases, the particle size increases. Here FeO-1

which is calcinated at 300°C has the crystalline size of about 22 nm and as the calcination temperature is increases to about 800°C for FeO-2, the crystallite size increases to 24 nm. Hence as the temperature increases the crystalline size also increases. Crystalline size and micro strain are interrelated and so the micro strain value of FeO-1 is high compared to FeO-2. Since microstrain and the defect parameter, dislocation density are correlated, the FeO-1 has more defects.

TABLE III. TEXTURE CO-EFFICIENT OF FEO-1

INTENSITY				
Observed	Standard	HKL	TC	
34.7	313	012	0.9536	
100	999	104	0.86102	
81.4	704	110	0.99456	
25.5	190	113	1.15443	
37.8	341	024	0.95349	
43.6	410	116	0.91471	
10.2	85	018	1.03219	
28.9	254	214	0.97869	
30.1	245	300	1.05677	
10.2	92	1 0 10	0.95366	
6.8	51	220	1.14688	

The preferential orientation of the films can be studied by calculating the texture coefficient TC(hkl) for all the planes. The variation in the texture co-efficient has been calculated for all the diffraction peaks i.e. the $(0\ 1\ 2)$, $(1\ 0\ 4)$, $(1\ 1\ 0)$, $(1\ 1\ 3)$, $(0\ 2\ 4)$, $(1\ 1\ 6)$, $(0\ 1\ 8)$, $(2\ 1\ 4)$, $(3\ 0\ 0)$, $(1\ 0\ 10)$ and $(2\ 2\ 0)$. It is observed that the $(1\ 1\ 3)$ plane has a high texture co-efficient of 1.15443. Hence the crystalline growth is mostly oriented in $(1\ 1\ 3)$ direction.

The variation in the texture co-efficient has been calculated for all the diffraction peaks of FeO-2 sample i.e. the (0 1 2), (1 USER © 2018 http://www.ijser.org 0 4), (1 1 0), (1 1 3), (0 2 4), (1 1 6), (0 1 8), (2 1 4), (3 0 0), (1 0 10) and (2 2 0). It is observed that the (2 2 0) plane has a high texture co-efficient of 1.76864.

INTENSITY		HKL	тс
OBSERVED	STANDARD		10
28.8	313	012	0.57636
100	999	104	0.62702
73.7	704	110	0.65576
26	190	113	0.85717
51.3	341	024	0.94235
63.1	410	116	0.96404
15.6	85	018	1.14962
45.4	254	214	1.11962
43.3	245	300	1.10706
18.1	92	1010	1.23236
14.4	51	220	1.76864

TABLE IV. TEXTURE CO-EFFICIENT OF FEO-2

4 CONCLUSION

The iron oxide nanoparticle with hematite phase was prepared using sol-gel process. Structural parameters were evaluated from the X-ray diffraction studies. From the evaluated structural parameters, the α -Fe2O3 nanoparticles prepared after calcination to a temperature of 800°C exhibited perfect lattice with less strain values.

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International Journal of Scientific & Engineering Research Volume 9, Issue 7, July-2018 ISSN 2229-5518

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